



Comparison of a Quantitative Diffusion-Trapping Model With Experiments on D Uptake in Damaged W

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- Introduction
- Model and Parameters
- Fluence and Temperature Variations
- Possible Mechanisms for Temperature-Dependent D Uptake
- Summary

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Introduction



- **Model system for benchmarking of diffusion-trapping code TESSIM [1]**
 - Polycrystalline tungsten damaged by MeV W ions
 - “Gentle” plasma loading
 - ↪ **Decoration** of traps with D **without further modification** of sample
 - **Analyse trapped D**
 - ↪ Nuclear reaction analysis (NRA): **depth profile**
 - ↪ Temperature programmed desorption (TPD): **binding energies**



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- Determine the remaining parameters **experimentally**
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- **Motivation: Huge discrepancy** of modelling and experiments at **Pilot-PSI [2]**
 - **Uptake** of **low-energy D** (at high plasma flux) surprisingly **slow**
 - Can only be reproduced by model if **incident flux** is **drastically reduced**
 - **Strong temperature dependence** of effective D uptake

Model and Parameters

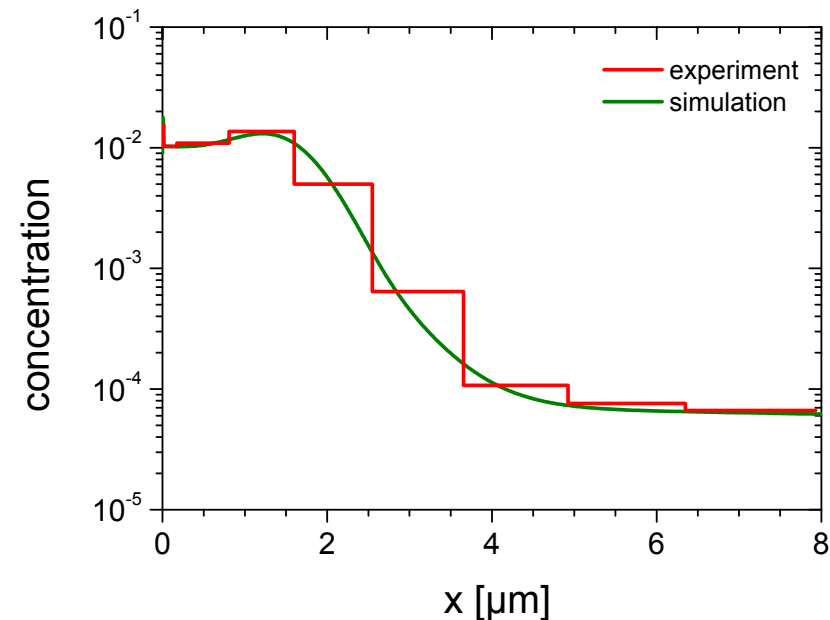
- **TESSIM** code: implemented in Mathematica by K. Schmid [1]
- Same basic principle as, e.g., TMAP7 [3]
 - Rate equations for **diffusion, trapping, implantation** and **surface loss**
- Highly **flexible** and **adaptive** code
 - Allows **arbitrary resolution** of **calculation grid**
 - Allows introducing any process that can be mathematically formulated (and is physically sensible), e.g.,
 - ↪ **Parametrised**, spatially distributed implantation source term
 - ↪ Trap densities evolving with time/fluence (e.g., blistering!)
- Good parallelisation by using Mathematica's NDSolve package
 - Currently using Method of Lines to solve PDE system



- Sample: **recrystallized W**, damaged by **20 MeV W⁶⁺** ions (**~0.5 dpa**)
 - **Very low background trap density**
 - W fluence sufficient to approximately **saturate radiation damage**
 - **No blistering** or strong “**surface peak**” observed
 - ↪ **Trap density** in damaged region appears **unaffected** by plasma exposure

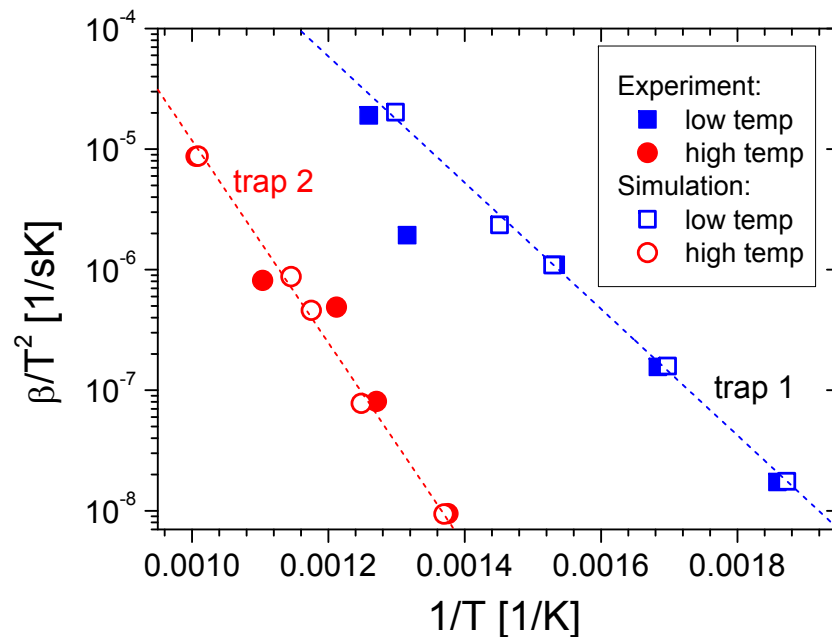


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 - ↪ **Trap density** in damaged region appears **unaffected** by plasma exposure
- **Reference** for simulations:
 - **3 days (72 h)** of plasma exposure at **floating potential** and **450 K**
- Total **trap density** replicates **NRA depth profile of D-saturated sample**

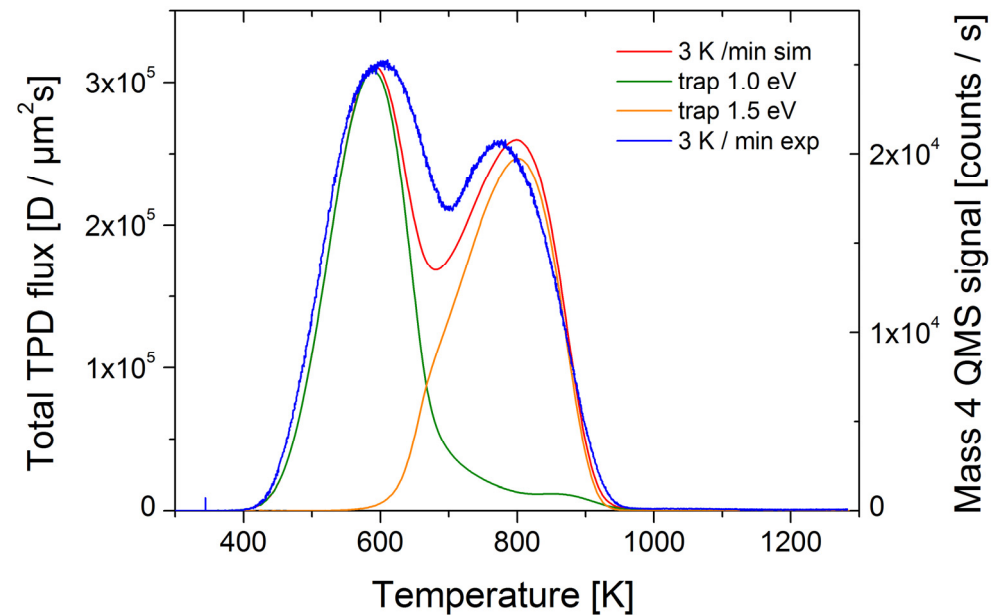
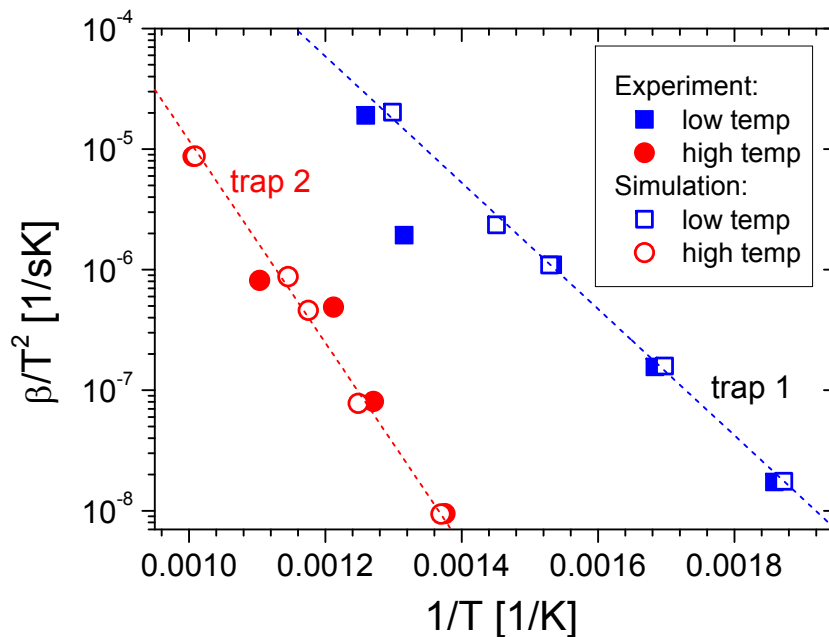




- Trap binding energies and pre-factors from TPD with ramp variations
 - Low energy trap: **1.0 eV, $1 \times 10^{10} \text{ s}^{-1}$** (damage peak and bulk)
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- **Density ratio** of traps replicates **TPD peak ratio**
 - **High energy trap** only in **damaged zone**
 - **Low energy trap** in **damaged zone** as well as in **bulk**





- **Incident flux: $5.6 - 9 \times 10^{19}$ D/m²s** (97% from D₃⁺, 2% from D₂⁺, 1% from D⁺)
 - Values from **characterisation** of plasma source **PlaQ** [4]
 - **Implantation profile**: parametrisation of **SDTrim.SP** calculations
 - ↪ For low ion energies as used here: **Gaussian fit** to profile is adequate
 - **Reflection yield: Eckstein's fit** for light projectiles [5]
 - Implantation profile and reflection yield **for each ion species**
 - Implemented as **functions of sample bias** voltage
- **Plasma potential 15 eV**, (negative) **bias voltage 0 V (floating) or 100 V**



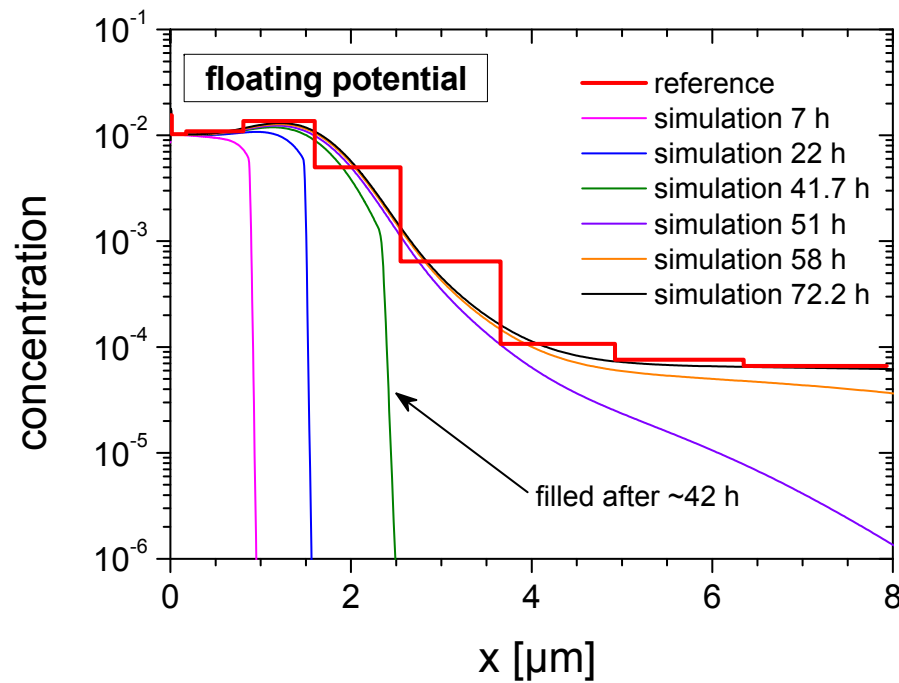
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- **Plasma potential 15 eV**, (negative) **bias voltage 0 V (floating) or 100 V**
- **Diffusion coefficient: Frauenfelder's** expression [6] **divided by $\sqrt{2}$** (for D!)
- **Assumption: Boundary condition** at surface: **$c = 0$**
 - **Surface loss of D is "infinitely" fast**
 - ↪ **Surface desorption** of D₂ (or D) is **not the rate-determining** step
 - ↪ Instead: determined by **transport** of D within W **to the surface**



Fluence Variation



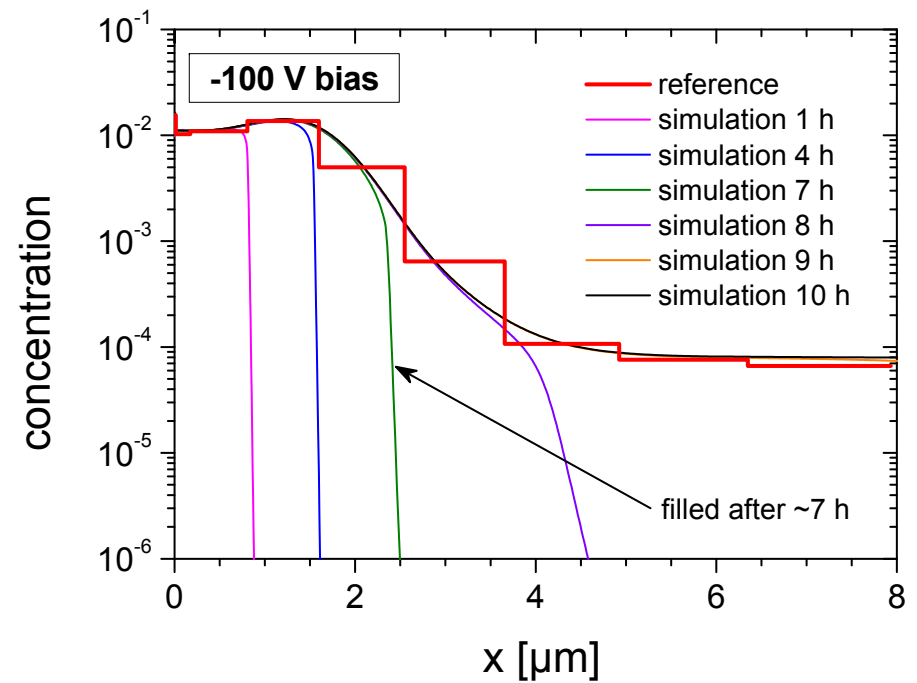
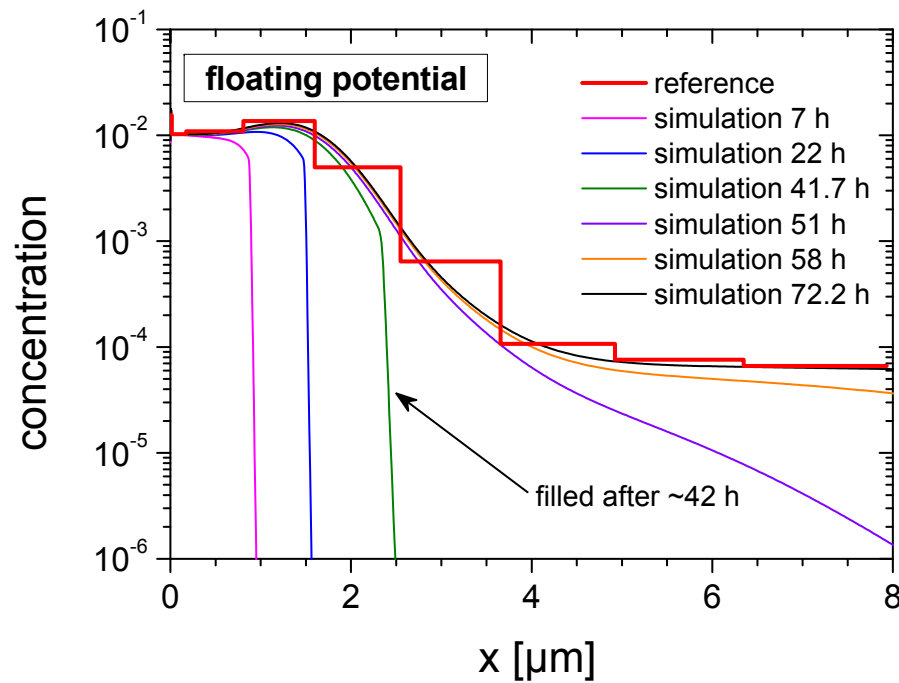
- Comparison: filling of damaged W (20 MeV W beam, 0.5 dpa) in PlaQ
 - without bias (~ 5 eV/D) at 450 K



- Damaged zone filled after **42 hours** without bias



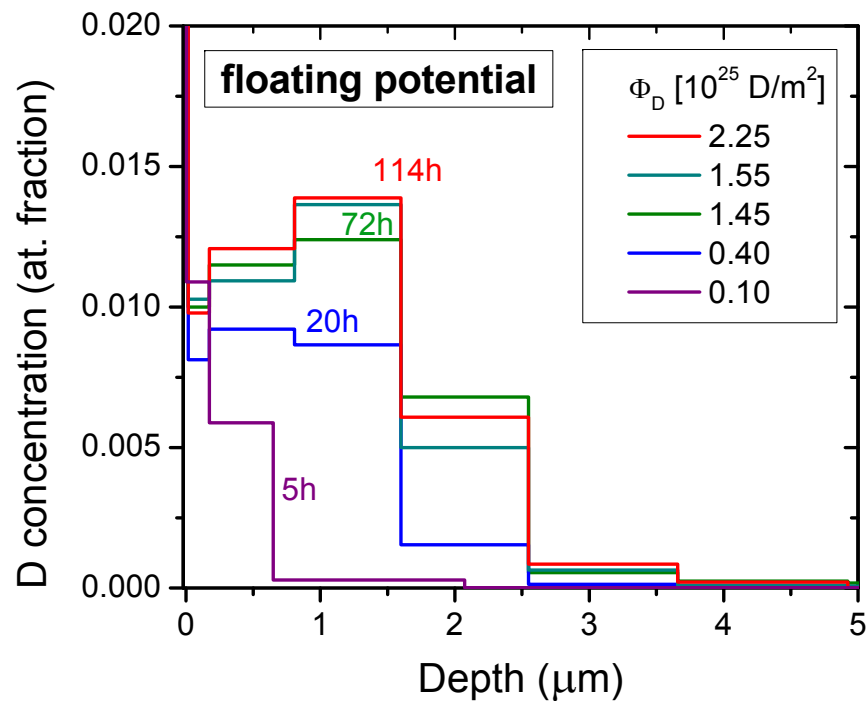
- Comparison: filling of damaged W (20 MeV W beam, 0.5 dpa) in PlaQ
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 - with -100 V bias (~ 38 eV/D) at 450 K



- Damaged zone filled after **42 hours** without bias, after **7 hours** with 100 V bias
 - Approximately **6x faster** filling with bias

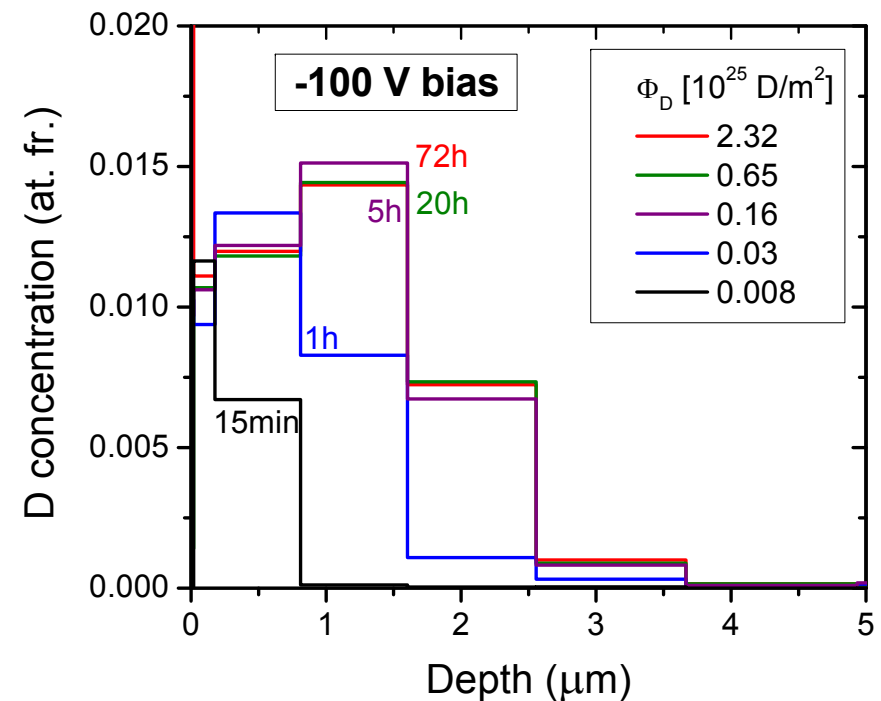
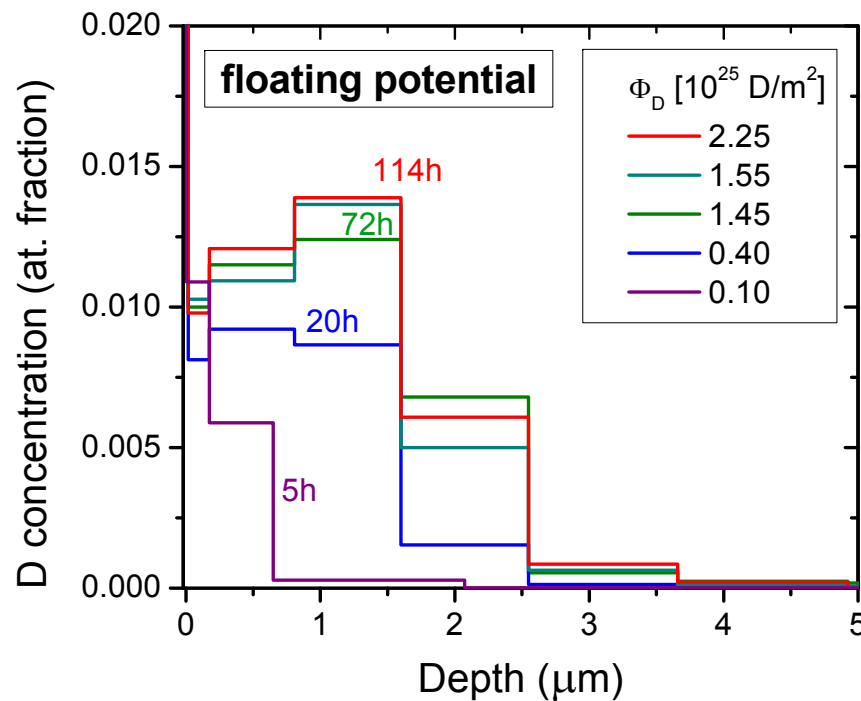


- **Experiment:** fluence variation at **450 K** sample temperature
 - **More than 20 h, less than 72 h** filling time at **floating potential**





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 - **More than 20 h, less than 72 h** filling time at **floating potential**
 - **More than 1 h, less than 5 h** at **-100 V bias**
 - **Closely matched by simulations**



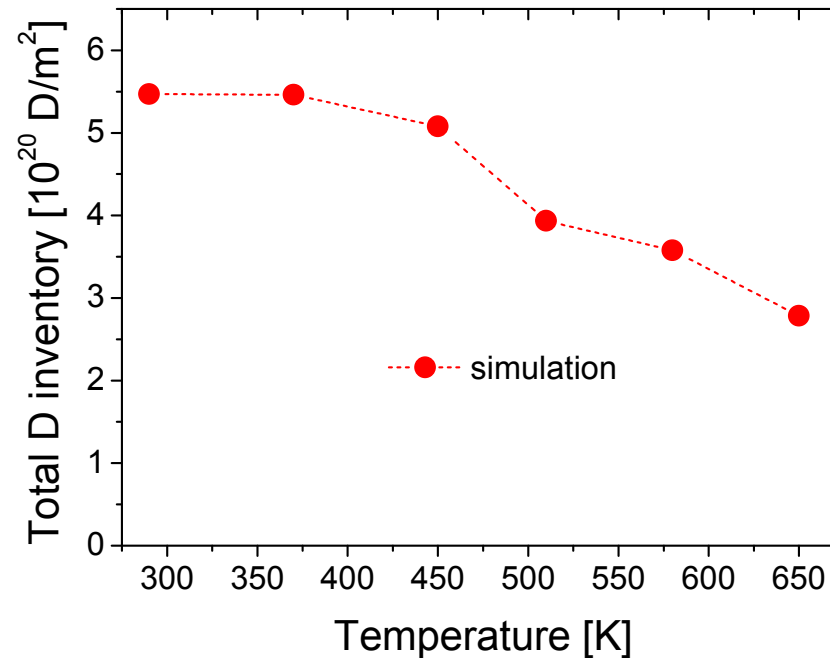


Temperature Variation



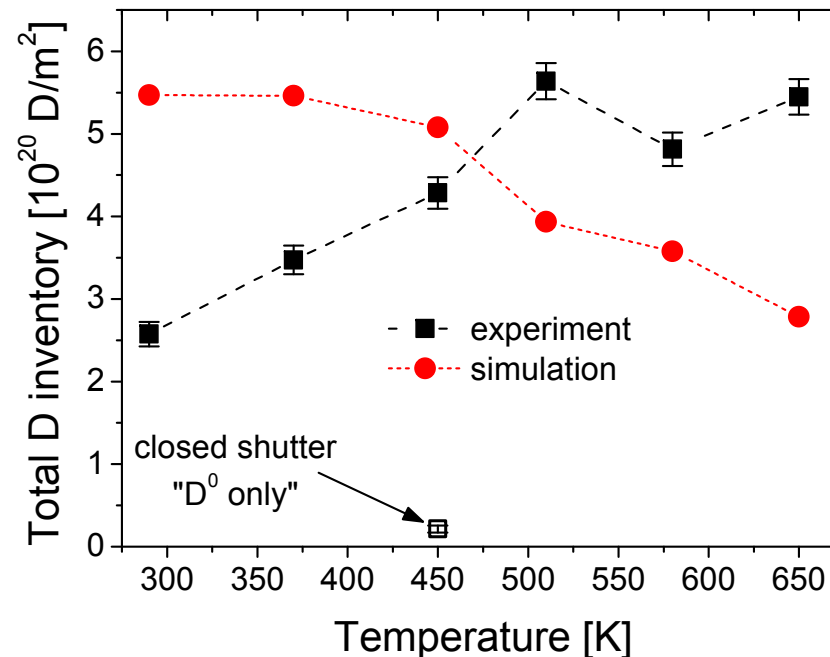
- Influence of **temperature** on filling of damaged W (**290 – 650 K**)
- Samples loaded with D at **floating potential** for **7 hours** each (**partially filled**)
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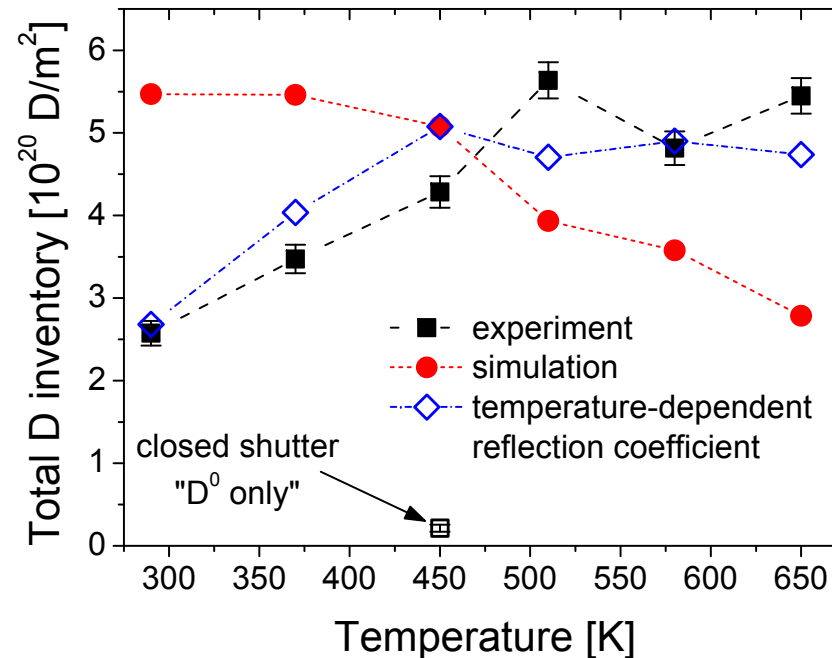
- Prediction: Inventory after 7 h nearly **independent** from **temperature** for \leq **450 K!**
 - **Rate-limiting step** is the **filling of traps** (not diffusion to traps)
 - **Higher temperatures: traps not fully populated** (detrapping!)

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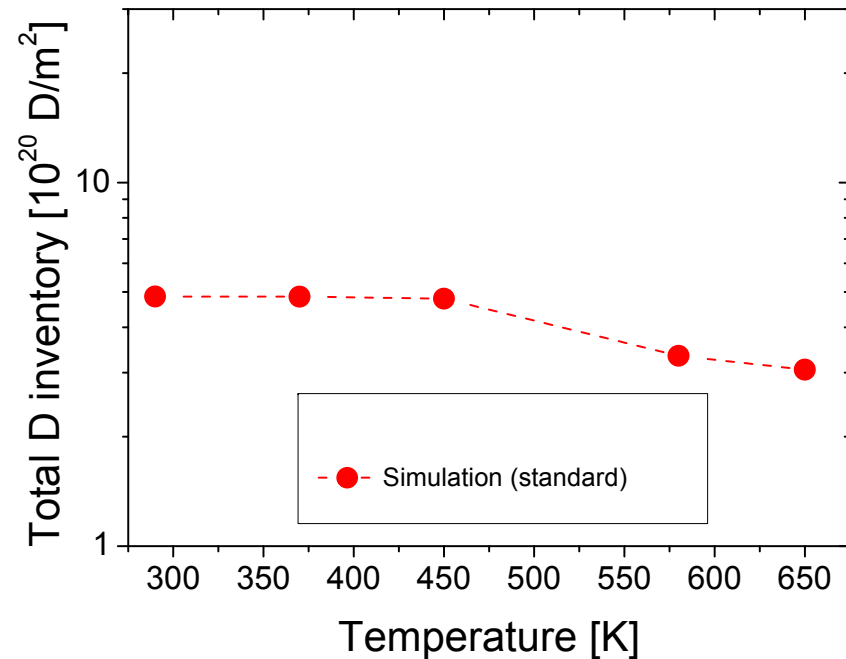
- **Experimental data:** total inventory **steadily increases** with temperature
 - **Close match at 450 K** → crossover point of trends
- **Trap densities** and **parameters fixed** by other experiments (450 K, TPD)
- Model results **insensitive** to **diffusion coefficient** (within 100x more or less)

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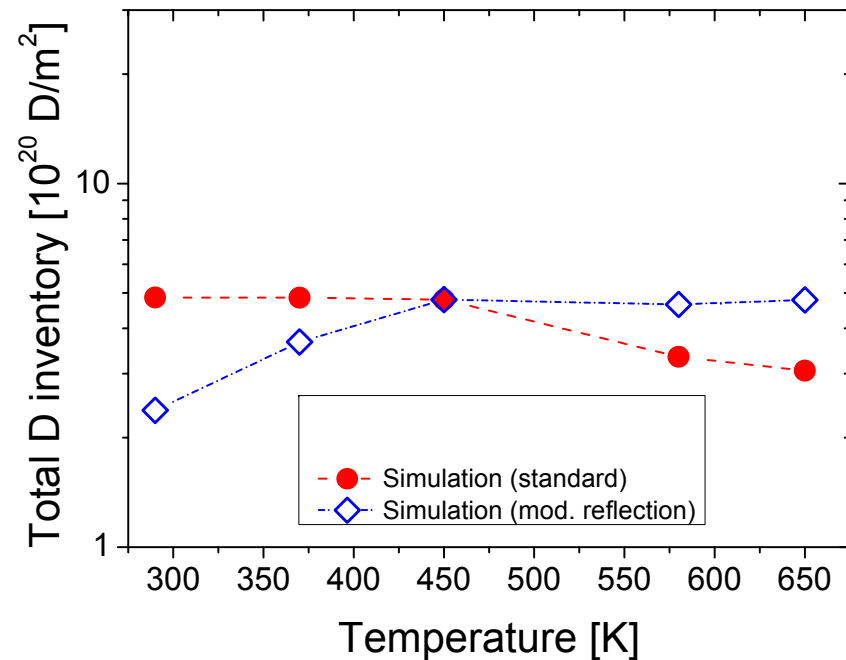
- Only parameter available to change results: **effective implantation flux**
 - Temperature-dependent D uptake with activation energy **~0.1 eV** → **match**
 - Same direction, but much **weaker dependence** than **t'Hoen et al. (1-2 eV)** [2]

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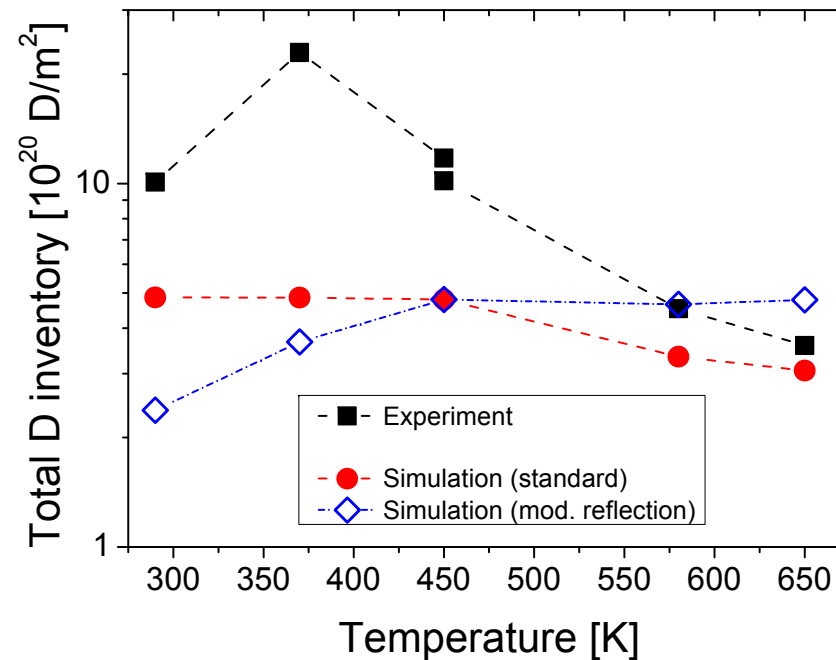


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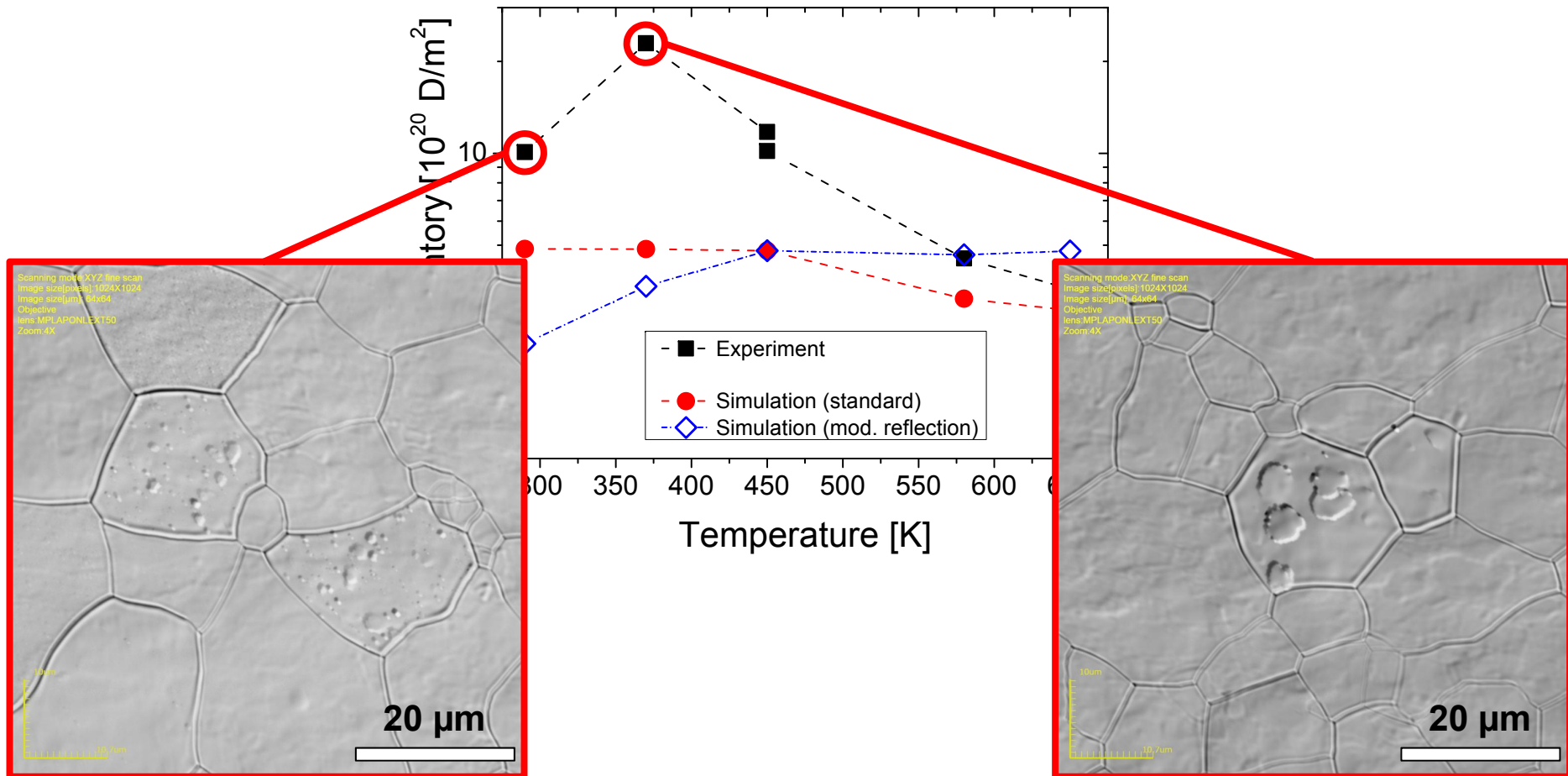




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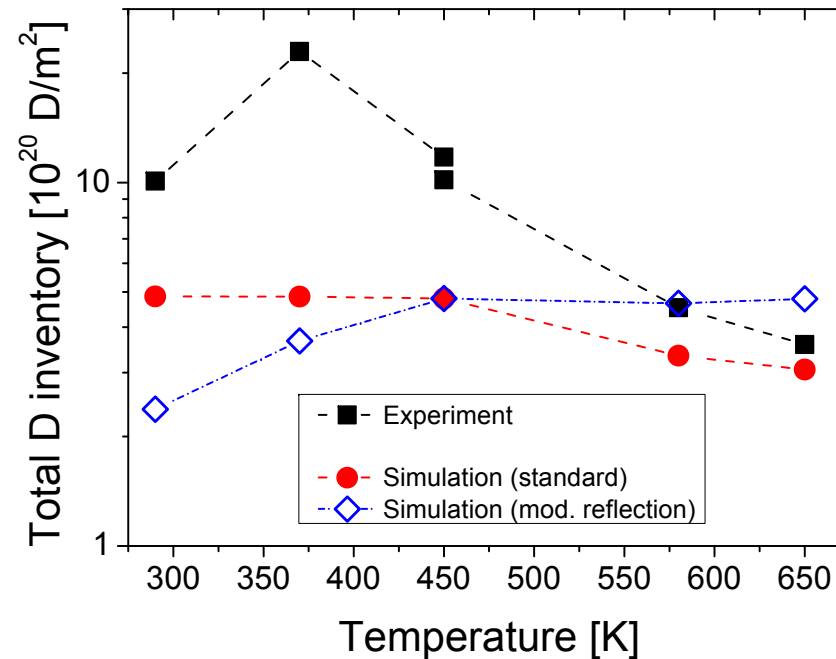


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- Samples **modified during plasma** exposure
 - Clearly visible **protrusions (“blisters”)** at 290 and 370 K
 - **No self-consistent model** for blistering currently available
 - **Not suitable** for elucidation of temperature dependence at floating potential

Mechanisms for Temperature Dependence



- **Temperature dependence of D uptake observed**
 - E.g., **here** and at **Pilot-PSI**
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 - Absorption of **atomic D** particularly at **higher temperatures**
 - **Enhanced desorption at high surface coverage** (i.e., low temperature):
 - ↪ But: **surface** already assumed to be **perfect sink** for H/D ($c = 0$ at $x = 0!$)
 - ↪ Relevant if actual reflection coefficient is much higher than assumed here



- Idea: **transport effect?**
 - Expected: $\Gamma_{\text{surface}} / \Gamma_{\text{bulk}} = r_{\text{implantation}} / r_{\text{diffusion}}$ (independent of temperature)
 - How can the **branching ratio** of **transport** towards surface and bulk **change**?
 - How can this effect be **stronger** for **lower temperatures**?



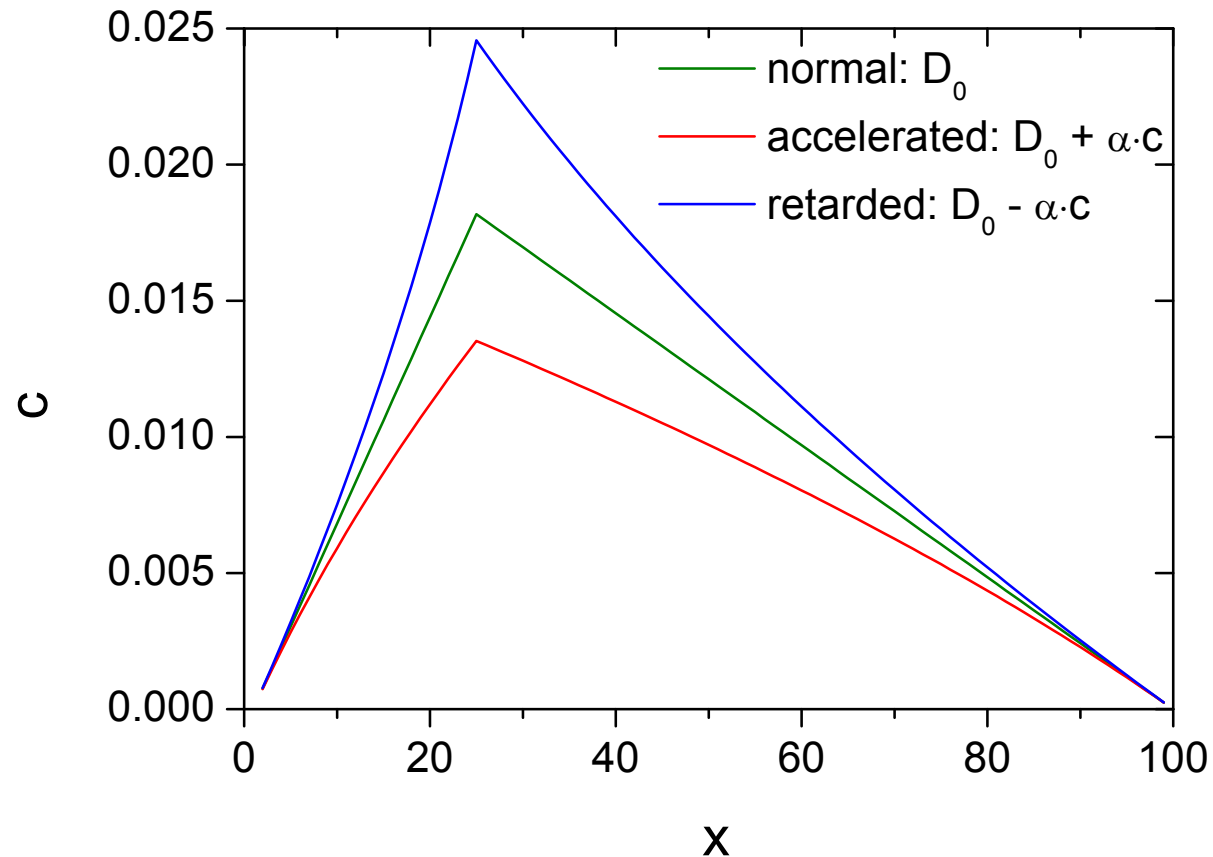
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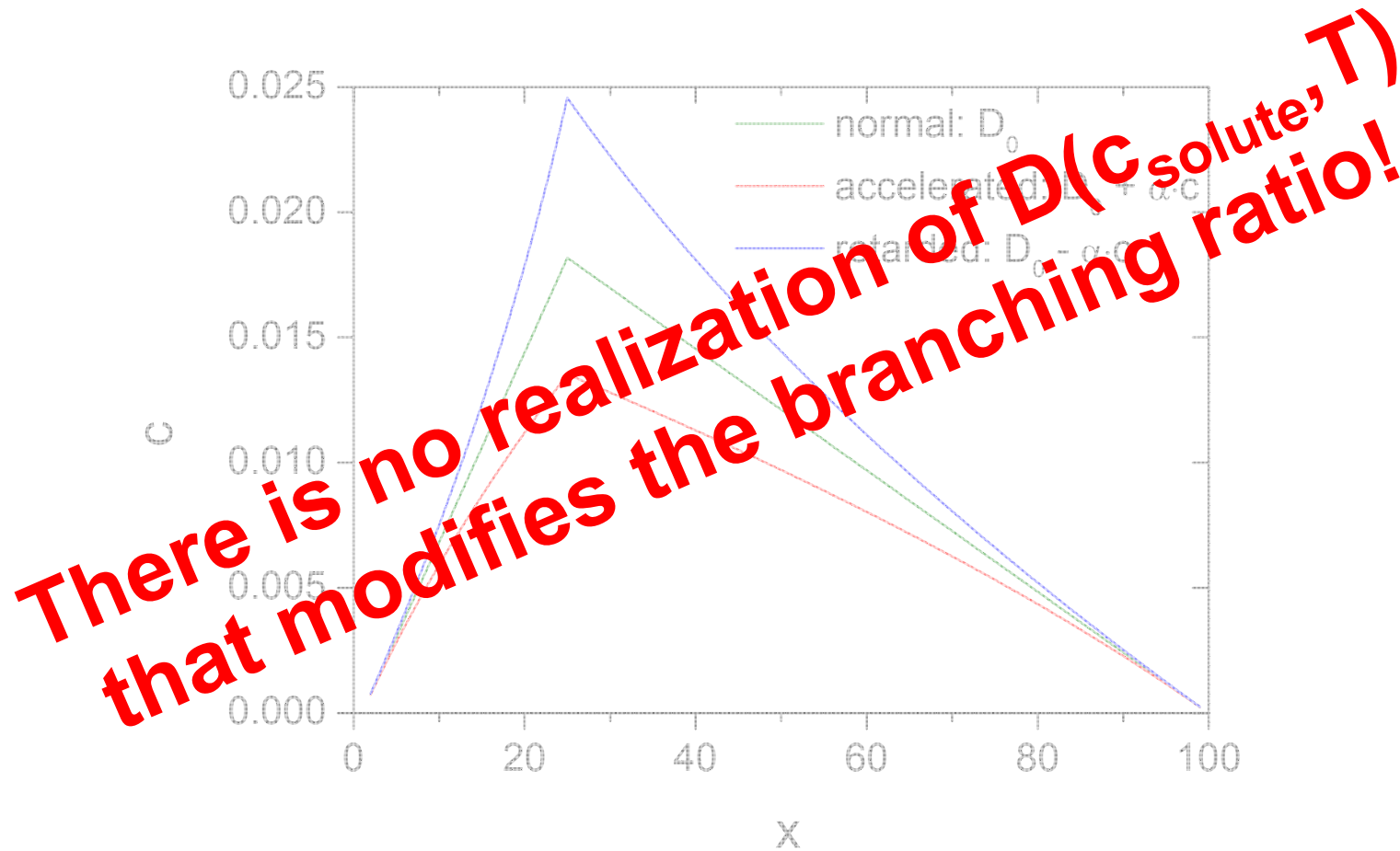


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- If diffusion equation becomes **non-local**, e.g., $D(T) \rightarrow D(c_{\text{solute}}, \nabla c_{\text{solute}}, T)$
 - Able to **change branching ratio**
 - **H-H interaction** (exception: not simple volume exclusion)
 - Example: **transport** depends on **stress** induced by **H concentration** (e.g., [9])



Summary



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 - Experiment and simulation show **similar range of values**, but **opposite trend**
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 - Results for **-100 V** bias **difficult to interpret** because of **blistering**
- **Possible mechanisms** for observed temperature dependence of D uptake
 - **Surface effect**
 - ↪ E.g., enhanced **reflection** or **re-emission**, or **D⁰ atom uptake**
 - **Non-local diffusion law**: e.g., **H-H interaction** leading to $D(c_{\text{solute}}, \nabla c_{\text{solute}}, T)$